

Bio-absorbable polymers in implantation-An overview

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Poly- α -hydroxy aliphatic esters are novel bio-absorbable polymers (BAPs), which are being used extensively as implantation products (orthopaedics, drug delivery, scaffolds and sutures). Polylactic acid (PLA), polyglycolic acid (PGA) and polydioxanone (PDO) are approved from food and drug administration agency (FDA) for human clinical uses. This review presents available synthetic routes for making bio-absorbable polymers, their properties and end use applications.

Keywords: Bio-absorbable polymer, Polylactic acid, Polyglycolic acid

Introduction

Orthopaedic surgeons have been repairing serious bone fractures by binding fractures with screws, pins, and other fixation-type devices using metals made up of highly sophisticated metal alloys of cobalt¹⁻⁴, titanium⁵⁻⁷, zirconium⁸⁻¹⁰, and tantalum¹¹⁻¹³. However, these metal alloys are hard and stiffer than bone and possibly interfere with regenerating bones. Search for more compatible materials with the human body led to consider bio-absorbable polymers (BAPs). In 1960s, Kulkarni *et al*^{14,15} implanted BAP as sutures and rod for repairing mandibular fractures in dog. Poly- α -hydroxyaliphatic esters are developed *in vivo* biomedical applications of orthopaedics, drug delivery systems, scaffolds, sutures and staplers. BAPs over alloys can eventually be resorbed or excreted by human metabolism without any side effects and exhibit more bone-like properties. Polylactic acid (PLA), polyglycolic acid (PGA) and polydioxanone (PDO) are some of the implant dominant BAPs (Scheme 1). Under ideal conditions, a BAP could encourage bone healing while body slowly metabolizes it, thus eliminating need for a second surgery that may be required when a metal alloy is implanted¹⁵⁻²⁰. Polymeric drug delivery devices prevent drug degradation and may also provide management of drug release by varying drug-to-polymer ratio, molecular weight and composition of

polymer^{21,22}. Bio-absorbable implants can be designed for fracture fixation, drug delivery, or ligament repair and other clinical use.

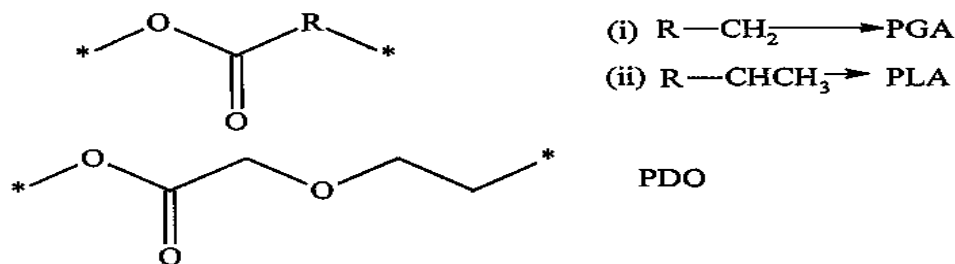
This review presents current issues on methods of preparation of bio-absorbable materials using various catalysts and initiators for making high molecular weight and functional BAPs, besides structure-property correlations of polymers prepared under various conditions and their end use applications.

Preparation of Bio-absorbable Polymers (BAPs)

Polycondensation

Under polymerization of lactic acid and glycolic acid via direct condensation, hydroxyl group present in \pm -position reduces reactivity of monomer and thereby increases reaction time. High purity is important to undergo effective reaction although commercially available monomer in the market is 85-90% pure²³⁻²⁵. Regular polycondensation method produces low molar mass products. Low molar mass oligo (lactic acid) (OLA), oligo (glycolic acid) (OGA) and combination of copolymers with functional monomers can be used to develop spherical microspheres for drug delivery systems and high molar mass polymers are also used for load bearing applications in biomedical implants²⁶⁻²⁸. Effect of catalytic action and method of polymerization plays an important role in synthesizing high molar mass polymers. Sn (II) and Ni (II) compounds show efficiency in synthesis of high molar weight polymers^{29,30}. Stannous octoate and tetraphenyltin catalysts have been approved by Food and

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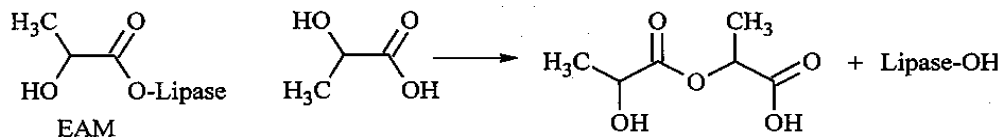
Scheme 1 — Repeating units of bio-absorbable polymers

Table. 1—List of catalyst and polycondensation conditions.

Monomer	Catalyst	Reaction Time h	Temp °C	Reduced pressure mmHg	Mw ($\times 10^4$)	PDI
LA ³²	Ti (BuO) ₄ -(0.1wt-%)	3/24	100/180	75/atm	0.5247	1.97
LA ³²	FeCl ₃ -(0.1wt-%)	3/24	100/180	75/atm	0.5226	1.64
LA ³²	H ₃ PO ₄ -(0.1wt-%)	3/24	100/180	75/atm	0.4518	1.58
LA ³²	AlCl ₃ -(0.1wt-%)	3/24	100/180	75/atm	0.2340	1.63
L-LA ⁵¹	TNBT	1/3/7/40	180	atm/760/ 760>1/1	13	3.30
L-LA ³³	Sc(OTf) ₃ (0.1 mol)	10/10/5	65/130/170	30/0.3/0.3	7.34	1.40
L-LA ³³	Sc(NTf ₂) ₃ (0.5 mol)	10/10/5	65/130/170	30/0.3/0.3	2.05	3.00
L-LA ²⁵	Tetraphenyltin	6/15	190	atm	4.3	3.58
L-LA ²⁵	Tetra-n-butylchlorodistannoxane	6/15	190	atm	2.47	3.85
D,L-LA ³⁴	SnCl ₂ ; pTSA (1:1-1wt %)	3/7	60/165	NS	9.116	1.90
D,L-LA ³⁴	Sn(Oct) ₂ ;pTSA(1:1-1wt %)	3/9	60/165	NS	9.464	2.00
OLLA ⁵⁵	GeO ₂ (0.8wt %)	20	180	10	2.8	-
OLLA ⁵⁵	Sb ₂ O ₃ (0.1wt %)	20	200	-	2.0	-
OLLA ⁵⁵	ZnO(0.1wt %)	20	200	-	3.6	-
OLLA ⁵⁵	Fe ₂ O ₃ (0.1wt %)	20	200	-	2.0	-
OLLA ⁵⁵	Al ₂ O ₃ (8.5wt %)	20	200	-	2.7	-
OLLA ⁵⁵	SiO ₂ (0.8 wt %)	20	180	-	1.1	-
OLLA ⁵⁵	TiO ₂ (0.8 wt %)	20	180	-	1.1	-
OLLA ⁵⁵	SnO (0.2 wt %)	20	180	-	5.0	-
OLLA ⁵⁵	SnCl ₂ .2H ₂ O (0.4 wt %)	20	180	-	4.1	-
OLLA ⁵⁵	TSA (0.34 wt %)	20	180	-	1.7	-
OLLA ³⁵	bmimAc(0.05 mol kg ⁻¹)	10	170	2.25	1.28	1.30
OLLA ³⁵	emimBr(0.05 mol kg ⁻¹)	10	170	2.25	1.04	1.17
OLLA ³⁵	bbimLLA(0.05 mol kg ⁻¹)	10	170	2.25	1.06	1.35
OLLA ³⁵	eeimLLA(0.05 mol kg ⁻¹)	10	170	2.25	1.25	1.34
OLLA ⁵⁴	SnCl/TSA(1:1-4wt%)	2/10	105/150	0/0.5	32	3.40
OGA ⁵⁰	Zn(CH ₃ CO ₂) ₂ .H ₂ O	1/4/20	190	150/30/atm	4.5	2.00

Drug Administration Agency (FDA). Protonic acids [methanesulfonic (MSA) and *p*-toluenesulfonic acids (*p*-TSA)] are found to be effective catalysts for esterification. Binary catalyst based on protonic acid and

tin compound yields high molar mass product in solid state condensation^{30,31}. Some of the synthetic parameters and catalytic systems result in high molar mass polymers (Table 1).



Scheme 2 — Enzyme activated polycondensation

Enzymatic polycondensation, which synthesizes polyester with easy removal of catalyst from polymer, is preferable for biomedical applications. Enzyme activates functional group of monomer and continues further condensation reaction with enzyme activated monomers (EAM) (Scheme 2). Origin of enzyme and feed ratios of monomers greatly influence microstructure of polymer and regioselective polymerization can be carried out under appropriate conditions^{36,37}. O'Hagan *et al*³⁸ and Binns *et al*³⁹ reported high molar mass aliphatic acid using lipase where byproduct (water) was removed by passing solvent through molecular sieves. Linko *et al*⁴⁰ prepared high molecular weight polyesters by removing water using vacuum. Uyama *et al*³⁷ proposed that regioselectivity is controlled to give linear polymers consisting of \pm , \acute{E} -disubstituted units at lower temperature, since enzyme activity shows poor conversion at higher temperature^{41,42}.

In chemical catalysis based solution condensation polymerization, rate is faster as boiling point of azeotropic solvent becomes higher due to continuous removal of water from reaction. This will enhance reaction since continuous recycling of solvent may carry some water into reaction and this may reduce molecular weight of polymers^{43,44}. Ajioka *et al*^{45,46} reported high molar mass (140 kDa) by drying solvent via molecular sieves and maintain water content (< 3ppm) for continuous recycling solvent incessantly during reaction (20-40h). Moon *et al*⁴⁷ and Fukushima *et al*⁴⁸ achieved high molar mass (500-600 kDa) by melt-solid condensation catalyzed by a $\text{SnCl}_2 \cdot 2\text{H}_2\text{O}/p\text{-TSA}$ binary system. In this condensation process, low molar mass is first prepared by traditional condensation, crystallized by heat treatment at 105°C for 1-2 h, and heated at 150°C for a further period of 30 h post solid condensation. Crystalline region increases in the system during heat treatment. At the same time, residual catalyst and dimers are transferred into amorphous phase. Further reaction of post solid condensation conducted in amorphous phase^{49,50}. Chen *et al*⁵¹ obtained high molar mass (130 kDa) via bulk polymerization at 180°C under reduced pressure using titanium(IV) butoxide (TNBT) as a catalyst. Nagahata *et al*⁵² proposed microwave assisted bulk condensation

using various catalysts and binary catalyst systems at 200°C under reduced pressure, resulting in moderate molar mass (16,000) with a broad PDI (3.5). In microwave assisted bulk condensation, reaction time can be shortened considerably when compared to conventional polycondensation⁵³. However, chain fragmentation and dimer formation gradually reduces molecular weight in bulk condensation and increases PDI at higher temperatures. Melt-solid condensation reaction carried out at $\leq 150^\circ\text{C}$, chain fragmentation have been controlled and high molecular weight polymers are obtained in high yield and in short reaction time, compared to bulk polymerization⁵⁴⁻⁵⁶.

Ring Opening Polymerization (ROP)

ROP of cyclic dimers has been examined to get high molecular mass and having specific micro structured polymers for biomedical applications using various metal initiators, organo metal complexes and metal free initiator systems. Carothers *et al*⁵⁷ reported that lactide can polymerize between 140-150°C and also observed that glycoside can polymerize with zinc chloride and can depolymerize into monomer under vacuum condition. ROP of cyclic dimers can be performed by four different mechanisms⁵⁸⁻⁶² (cationic, anionic, coordination-insertion and enzymatic polymerization). Cuiping *et al*⁶³ studied reactivity of rare earth initiators using lactide at 90°C in toluene and summarized catalytic efficiency as: $\text{La} > \text{Gd} > \text{Nd} > \text{Y} > \text{Er}$ ($M_n \leq 54$ kDa).

Magnesium and zinc catalysts form M^{2+} salts with very similar ionic radii and these metals are essential for life. Mg, Zn, Li, Ca, Fe and lanthanides are biocompatible metal initiators and preferred catalysts for biomedical applications⁶⁴⁻⁷². Heavy metal catalysts (Cd, Sn, Pb, Zr, Bi and Y) are not recommended for biomedical applications. Although Sn(II) (2-ethylhexanoate) is a general catalyst used to high molecular weight because of solubility in organic solvent, monomer and in melt state polymer. Sn(II) (2-ethylhexanoate) produces optically pure polylactide for biomedical applications and residual heavy metal ions are removed by post purification process^{72,73}.

Table 2—List of initiator and ROP condition

Monomer	Initiator	T °C	Time, h	η_{inh}	Mnx10 ³	PDI	Solvent
D,L-lactide ⁷⁹	Y(OCH ₂ CH ₂ OiPr) ₃	RM	3.000	-	14.5	1.23	DCM
L-lactide ⁸⁰	Al(OTf) ₃	100	6.000	-	4.30	1.38	In Air
L-lactide ⁸⁰	In(OTf) ₃	100	6.000	-	3.50	1.12	In Air
L-lactide ⁸⁰	Sc(OTf) ₃	100	24.00	-	0.84	3.77	In Air
L-lactide ⁸⁰	Cu(OTf) ₃	100	24.00	-	0.80	2.76	In Air
L-lactide ⁸⁰	AgOTf	100	240.0	-	0.58	2.19	In Air
L-lactide ⁸⁰	Mg(OTf) ₃	100	240.0	-	0.76	2.45	In Air
L-lactide ⁸⁰	Y(OTf) ₃	100	240.0	-	0.94	3.18	In Air
L-lactide ⁸⁰	La(OTf) ₃	100	240.0	-	0.90	2.57	In Air
L-lactide ⁸⁰	Sm(OTf) ₃	100	240.0	-	0.96	3.86	In Air
L-lactide ⁸⁰	Yb(OTf) ₃	100	240.0	-	1.16	2.94	In Air
D,L-Lactide ⁸¹	(R)-SALBinaphtAlOCH ₃	70	281.0	-	12.7	1.27	Tol
L-lactide ⁸²	[5-Cl-salen]AlOMe	RM	120.0	-	12.0	1.10	DCM
D,L-Lactide ⁸²	[5-Cl-salen]AlOMe	RM	120.0	-	12.0	1.10	DCM
rac-Lactide ⁸³	Aluminum/Schiff base initiator	70	8.950	-	22.3	1.04	Tol
rac-Lactide ⁸⁴	BuLi	20	0.667	-	45.0	1.60	THF
rac-Lactide ⁸⁴	Bu ₂ Mg	20	0.667	-	23.0	1.50	THF
D,L-lactide ⁸⁵	[(MeC(O)CHC(NCH ₂ -CH ₂ O)Me)AlCl ₂ /CHO	70	18.00	-	15.86	1.15	Tol
L-Lactide ⁸⁶	Bu ₂ Mg	0	96.00	3.01	-	-	Tol + CE
L-Lactide ⁸⁶	BuMgCl	80	86.00	0.45	-	-	Tol
D,L-Lactide ⁸⁷	La	90	0.833	-	54.3	1.64	Tol
D,L-Lactide ⁸⁷	Gd	90	0.833	-	41.2	1.60	Tol
D,L-Lactide ⁸⁷	Nd	90	0.833	-	25.8	1.55	Tol
D,L-Lactide ⁸⁷	Y	90	0.833	-	8.90	1.41	Tol
L-Lactide ⁸⁸	Calcium dimethoxide	120	1.500	-	7.30	1.43	Bulk
L-Lactide ⁸⁹	Sn alkoxide	60	163.0	-	116	1.18	Tol
L-Lactide ⁹⁰	Creatinine	165	96.00	-	14.0	1.30	Bulk
L-Lactide ⁹¹	Dibutyltin Dimethoxide	70	20.00	-	40.1	1.65	Tol
L-Lactide ⁹²	FeCl ₂	150	192.0	0.40	-	-	Bulk
L-Lactide ⁹²	FeGlyC ₂	150	192.0	0.5	-	-	Bulk
L-Lactide ⁹²	FeLaC ₂	150	192.0	0.72	-	-	Bulk
L-Lactide ⁹²	FeMand ₂	150	192.0	0.38	-	-	Bulk
L-Lactide ⁹²	SnOct ₂	120	96.00	2.65	-	-	Bulk
rac-lactide ⁹³	2,2-Diethyl-propane-1,3-diamine	70	5.616	-	11.7	1.08	Tol
D,L-Lactide ⁹⁴	Fe(OEt) ₃	130	48.00	-	61.4	1.66	Bulk
D,L-Lactide ⁹⁴	Fe(OPr) ₃	130	48.00	-	50.8	1.68	Bulk
D,L-Lactide ⁹⁴	Fe(O ⁱ Pr) ₃	130	48.00	-	33.4	1.73	Bulk
D,L-Lactide ⁹⁴	Fe(OBu) ₃	130	48.00	-	20.7	1.92	Bulk
L-Lactide ⁹⁵	C ₁₅ H ₂₅ AgClN ₃ O ₂	160	4.000	-	5.10	1.23	Bulk
L-Lactide ⁹⁵	C ₁₅ H ₂₄ AuClN ₃ O ₂	160	4.000	-	5.40	1.17	Bulk
L-Lactide ⁹⁶	Ge	120	162.0	-	41.4	1.40	CB
L-Lactide ⁹⁷	HBG. OAc	120	18.00	-	20.5	1.07	Bulk
D,L-Lactide ⁹⁷	HBG. OAc	120	18.00	-	20.0	1.10	Bulk
rac-Lactide ⁹⁸	[(THF)NaFe(O ⁱ Bu) ₃] ₂	RT	7.00	-	61.5	1.49	CH ₂ Cl ₂
D,L-Lactide ⁹⁹	La(OiPr) ₃	21	0.500	-	21.9	1.20	DCM/Tol
D,L-Lactide ¹⁰⁰	Zn	140	21.30	-	9.5	1.40	Bulk

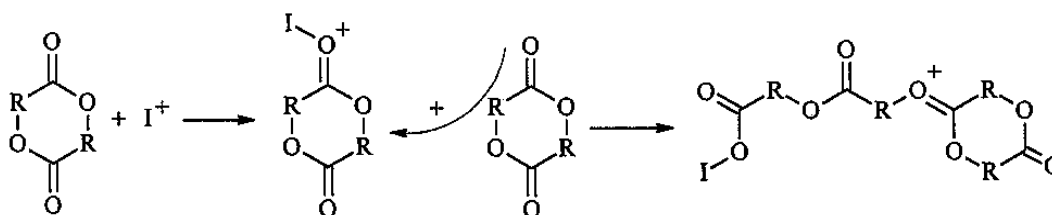
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D,L-Lactide ¹⁰⁰	Zn-lactate	140	11.90	-	105	1.70	Bulk
L-Lactide ¹⁰¹	Zn(II) L-lactate	150	96.00	0.9	-	-	Bulk
L-Lactide ¹⁰¹	Zn(II) L-mandelate	150	96.00	0.61	-	-	Bulk
L-Lactide ¹⁰¹	Zn(II) stearate	150	144.0	0.68	-	-	Bulk
L-Lactide ¹⁰¹	ZnBr ₂	150	192.0	0.89	-	-	Bulk
L-Lactide ¹⁰¹	ZnI ₂	120	192.0	0.60	-	-	Bulk
L-Lactide ¹⁰²	Zn aceturate	150	192.0	0.65	-	-	Bulk
L-Lactide ¹⁰²	Zn prolinatate	150	192.0	0.62	-	-	Bulk
L-Lactide ¹⁰²	Zn pyroglutamate	150	48.00	0.53	-	-	Bulk
L-Lactide ¹⁰²	Zn tosylglycinate	150	96.00	0.52	-	-	Bulk
L-Lactide ¹⁰²	Zinc <i>N</i> -acetyl-4-aminobenzoate	150	48.00	0.59	-	-	Bulk
Gl: La: Cap ¹⁰³	Zirconium(IV) acetylacetonate	100	-	1.70	62.6	1.70	Bulk
L-Lactide ¹⁰⁴	Ytrocene complexes	25	70.00	-	44	1.20	CH ₂ Cl ₂
D,L-Lactide ¹⁰⁴	Yttrium(III) Complexes	25	24.00	-	62.5	2.13	CH ₂ Cl ₂
D,L-lactide ¹⁰⁵	Tetra(phenylethynyl)tin	0.25	140.0	-	599.9	1.91	Bulk
L-Lactide ¹⁰⁶	Sn(OBu) ₂	19	120.0	-	968	1.36	Bulk
D,L-lactide ¹⁰⁷	2,6-dimethylaryloxyde	100	0.750	-	36.1	1.59	Tol
L-Lactide ¹⁰⁸	Titanium Alkoxide Complex	130	12.00	-	15	2.33	Bulk
Lactide ¹⁰⁹	Tetra(phenylethynyl)tin	140	0.25	-	599.9	1.91	Buk
L-Lactide ¹¹⁰	[{1-Isopropyl-3-(<i>N</i> -phenylacetamido)imidazol-2-ylidene}+ Cl	100	4.000	-	13.5	1.23	Bulk
D,L-Lactide ¹¹¹	[(Me ₃ TAC) ₂ Pr(OTf) ₃]	170	18.00	-	1.8 ^{Ml}	-	Bulk
D,L-Lactide ¹¹²	[(Me ₃ SiNCH ₂ CH ₂) ₂ NMe]AlH	80	160.0	-	16.93	1.72	Ben
D,L-Lactide ¹¹²	{[(Me ₃ SiNHCH ₂ CH ₂)(Me ₃ SiNCH ₂ CH ₂)NMe]AlCl}{AlCl ₄ }/PO	80	600.0	-	18.50	1.39	Ben
D,L-Lactide ¹¹³	(CH ₃ COO) ₃ La	140	24.00	-	54.8 ^{Ml}	-	Bulk
D,L-Lactide ¹¹³	(CCl ₃ COO) ₃ Y	160	30.00	-	59 ^{Ml}	-	Bulk
D,L-Lactide ¹¹³	(CCl ₃ COO) ₃ La	160	1.000	-	72 ^{Ml}	-	Bulk
D,L-Lactide ¹¹³	(CCl ₃ COO) ₃ Sm	160	10.00	-	69 ^{Ml}	-	Bulk
D,L-Lactide ¹¹³	(CCl ₃ COO) ₃ Nd	160	11.00	-	71 ^{Ml}	-	Bulk
L-Lactide ¹¹⁴	Mg/BnOH	83	2.000	-	45.2	1.21	C ₂ H ₄ Cl ₂
D,L-Lactide ¹¹⁵	Sn(O <i>i</i> Pr) ₄	70	8.500	-	36.7	1.29	Tol
D,L-Lactide ¹¹⁵	Sn(Oct) ₂	70	8.000	-	34.2	1.05	Tol
D,L-Lactide ¹¹⁵	Sn(O <i>t</i> Am) ₄	70	8.500	-	43.6	1.47	Tol
<i>rac</i> -lactide ¹¹⁶	Sn- <i>rac</i> -LA	80	2.000	-	30.8	1.2	Tol
<i>rac</i> -lactide ¹¹⁶	Sn-CL	130	2.000	-	430.1	1.3	Bulk
L-Lactide ¹¹⁷	[(SalenMe)Mg(OBn)] ₂	25	0.750	-	27.3	1.09	Tol
L-Lactide ¹¹⁷	[(SalenMe)Zn(OBn)] ₂	25	4.500	-	34.1	1.03	Tol
D,L-Lactide ¹¹⁸	[(L ^{Me2})(HL ^{Me2})Y ₃ (OCH ₂ C ₆ H ₅) ₄]	25	16.00	-	14.5	1.23	CH ₂ Cl ₂
L-Lactide ¹¹⁹	Ti(OP <i>r</i> i) ₄	130	2.000	-	10.54	1.90	Tol
L-Lactide ¹²⁰	Bu ₂ SnOct ₂	120	24.00	-	0.52	-	Bulk
L-Lactide ¹²¹	Red-Al	110	48.00	-	23	1.12	Tol
<i>rac</i> -Lactide ¹²²	(MeMgOEt ₂)bamam-1	RT	2.500	-	61.1	1.44	CDCl ₃
L-Lactide ¹²³	Iron Isobutyrate	190	2.000	-	148	1.60	Bulk
L-Lactide ¹²⁴	Strontium Isopropoxide	80	3.000	0.789	45.4	2.26	Tol
L,L-Lactide ¹²⁵	Sn(Oct) ₂	110	50.00	-	304	2.25	Bulk

Contd.

L-Lactide ¹²⁶	La(OTBP) ₃	100	0.333	-	46.9	1.49	Tol
D,L-lactide ¹²⁷	K[CuL]2H ₂ O	130	24.00	-	4.86 ^{Ml}	-	Bulk
D,L-lactide ¹²⁷	K[ZnL]2H ₂ O	130	24.00	-	5.23 ^{Ml}	-	Bulk
D,L-lactide ¹²⁷	K[CoL]3H ₂ O	130	24.00	-	5.26 ^{Ml}	-	Bulk
D,L-lactide ¹²⁷	K[NiL]3.5H ₂ O	130	24.00	-	11.1 ^{Ml}	-	Bulk
<i>rac</i> -Lactide ¹²⁸	[(R)-(SalBinap)AlOiPr] [(R)-1]	70	40.00	-	22.6	1.09	Tol
L-Lactide ¹²⁹	Zn pyroglutamate	150	48.00	0.53	-	-	Bulk
L-Lactide ¹²⁹	Zn tosglycinate	150	96.00	0.52	-	-	Bulk
L-Lactide ¹³⁰	(BDI-OMe)Zn(μ-OBn) ₂ Zn (μ-OBn) ₂	25	0.166	-	29	1.06	Bulk

^{Mn} -Viscosity average molecular weight, Tol-Toluene, THF-tetrahydrofluorine, DCM-Dichloromethane, Ben-Benzene, CE-Crown ethers, CB- Chlorobenzene, PO-Propylene oxide, GI-glycolide, La- lactide, Cap- ε-caprolactone, PDI-Polydispersity index



Scheme 3 — Cationic polymerization mechanism

Microwave assisted ROP (MROP) with organo metallic catalyst is an appealing route for ring opening due to its high efficiency and uniform heat distribution during the reaction. ROP and polycondensation reactions are very impressively accelerated by microwave irradiation^{74,75}. Zhang *et al*⁷⁶ achieved molecular mass of 10⁵ g/mol using D,L-lactide with microwave energy (255W) for 10 min. However, higher microwave energy with increasing time causes chain scission and reduces molecular mass to oligomers and cyclic dimers. Li *et al*⁷⁷ observed that MROP shows higher rate of conversion and high molecular mass product compared with conventional ROP.

Pure D and L forms of lactic acid monomer produce isotactic polylactic acid. Syndiotactic polylactic acid was produced by *meso*-lactide. Polymerization occurs without any stereoregularity led to heterotactic polylactide. Rate of bio-absorption or degradation rate, mechanical and thermal properties depends on microstructural arrangement of polymers⁷⁸. Recently, many researchers have carried out ROP of lactide and other bio-absorbable polymers (Table 2).

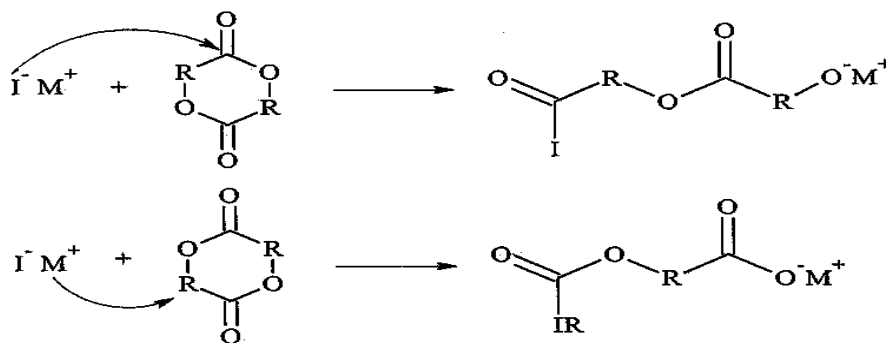
Cationic Polymerization

Cationic ROP involves extremely strong acids or carbenium ion donors, which are capable of initiating a

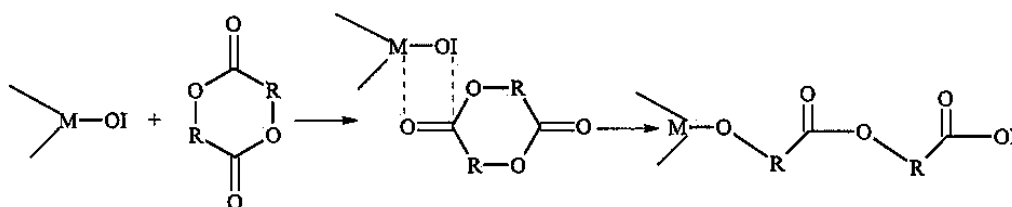
cationic polymerization of cyclic dimers. Formation of positively charged species and subsequent attack by monomer results in ring opening of positively charged species^{130,131} (Scheme 3). Kricheldorf *et al*¹³² reported that cationic polymerization is difficult to control at higher temperature, which tends to more or less racemization and this affect properties of polymers rather slowly under 50°C. Among numerous acidic compounds investigated, only trifluoromethane sulfonic acid (HOTf) and methyl trifluoromethane sulfonate (MeOTf) proved to be efficient initiators. Polymerization rates were significantly higher in nitrobenzene than in chlorinated solvents, and 50°C being optimum reaction temperature. But, lower temperatures result in modest yields and higher temperatures lead to dark-coloured samples. Optical rotation measurements revealed that samples of 100% optically active poly(L-lactide) are obtained from L-lactide using HOTf and MeOTf^{133, 134}.

Anionic Polymerization

Under anionic polymerization, nucleophilic attack of a negatively charged initiator on carbonyl group or on carbon atom adjacent to acyl-oxygen results in ring opened linear polyester (Scheme 4). Lactide ions can initiate a new chain and thereby deprotonation of monomer leads to chain transfer process. Kricheldorf



Scheme 4 — Anionic polymerization mechanism



Scheme 5 — Coordination insertion mechanism

*et al*¹³⁵ proposed bulk polymerization of lactide using FeLac_2 at 150°C and observed that FeLac_2 caused intensive racemization of monomers. But, ZnI_2 yield a moderate molar mass (optical purity, 97%) even at $120\text{-}150^\circ\text{C}$ ¹³⁶. Ionic chain ends in anionic and cationic polymerisation of lactones causes transesterification reactions even at moderate temperatures and results in cyclization. Imidazole and *N*-methylimidazole catalyzed lactide polymerization gave low molar mass cyclic polymer at 100°C ¹³⁷.

Coordination Insertion ROP

Coordination insertion ROP mechanism enhances electrophilicity of CO-group and nucleophilicity of OR-groups so that an “insertion” of lactone into metal O-bond may occur¹³⁸ (Scheme 5). Propagating chain remains attached to metal through an alkoxide bond during propagation. Reaction is terminated by hydrolysis forming a hydroxyl end-group. Macromers with active end-groups are produced in post polymerization reactions of functional alkoxy substituted initiators¹³⁹.

Enzymatic Ring Opening Polymerization

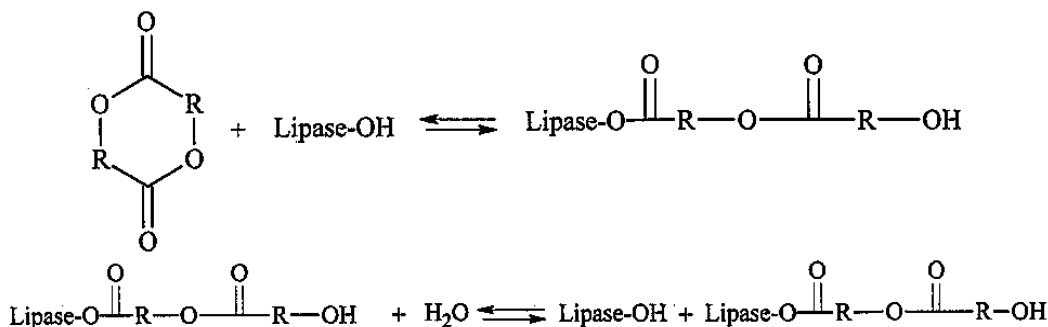
Enzymatic polymerization can be carried out using low temperature by enzyme activated monomers

(EAM)^{140,141}. Lipase catalyzed polymerization is an eco-friendly technique for preparation of useful polyesters by polycondensation as well as ring opening polymerization (Scheme 6). Enzyme-catalyzed reactions accelerate chain degradation and backbiting at high temperatures^{142,143}.

Properties of Bio-absorbable Polymers

Thermal Properties

Glass transition temperature (T_g) of important BAP's is as follows: PLA, $55\text{-}60$; PGA, 35 ; and PDO, -10°C . Melting temperature (T_m) is as follows: PLA, $145\text{-}180$; and PGA, $220\text{-}230^\circ\text{C}$. After molding, internal stress tends to generate micro cracks, affecting overall performance of polymer¹⁴⁴⁻¹⁴⁶. Internal stress can be removed by annealing material for 4 h at 120°C after molding. Lamellar crystallites in poly(L-lactide) appears to be very stable towards hydrolysis¹⁴⁷. Generally, PLA exhibits left-handed helix conformation for α -form. Fibers can be drawn in the case of high molar mass PLLA, which can change to stable β -form¹⁴⁸⁻¹⁵⁰. PGA has better thermal stability than PLA. Thermal degradation of PLA and PGA in random chain scission shows activation energies of 16.5 and 10.2 kcal/mol and in specific chain end scission, it absorbed 28 and 22.2



Scheme 6 — Enzymatic ring opening polymerization

kcal mol respectively¹⁵¹. Thermal degradation is very fast in presence of moisture because PLA, PGA and PDO are moisture and heat sensitive. Degradation that occurs under hydrolysis in presence of a trace amount of water, zipper-like depolymerization due to shearing on processing, random main chain scission by oxidation and intermolecular transesterification results in low molar mass product^{152,153}.

Mechanical Properties

PLA, PGA and PDO exhibit good mechanical properties among all synthetic BAP's. PGA and PLA have tensile strength usually 40-140 MPa. Elastic modulus is as follows: PLA, 2.7; PGA, 7.0; and PDO, 1.5 GPa. PGA shows higher elongation and tensile modulus than PLA¹⁵⁴⁻¹⁵⁸. These values are lower than those of cortical bone^{159,160}, which has: bending strength, 180-195 MPa; shear strength, 68 MPa; and elastic modulus, 6-20 GPa. Good mechanical strength of BAP is attainable if material is self-reinforced (SR) by sintering^{161,162}. SR polymeric composites significantly increase strength, ductility and elastic modulus. Composite structure of SR polymer made up of polymeric material comprised of oriented reinforcing units and a binding matrix. Both matrix and reinforcing materials are in crystalline nature and high degree of similar molecular orientation makes SR polymer implants to be stiff and strong in oriented direction¹⁵⁴. For SR-PLLA pins, an initial bending strength of 300 MPa and shear strength of 220 MPa needed¹⁶¹. For SR copolymers, bending strength and Initial shear strength values, respectively, are as follows: SR-Poly(L-DL)LA (70:30), 214 MPa¹⁶², 121 MPa¹⁶²; and SR-PLA/PGA (80:20), 226 MPa¹⁶³, 115 MPa¹⁵⁰. Elastic modulus of SR pins is 6-10 GPa^{154,161-163} being close to that of cortical bone. New fabrication technologies have been developed to process BAPs into tissue engineering scaffolds¹⁶⁴⁻¹⁶⁶.

Bio-absorbable Properties

Biodegradation¹⁵¹⁻¹⁵⁴ mainly depends on type of material, molar mass, percentage of crystallinity, type of environment (enzymatic concentration, external stress) and material molding history (internal stress). BAP has been accomplished with hydrolytically unstable linkages in backbone, which tend to lose molar mass after implantation. Material undergoes specific and/or non-specific degradation *in vivo*¹⁵⁵⁻¹⁵⁷. PLA and PGA produces acid product during degradation. PDO has manifested into nontoxic material on decomposition. Toxic effect can occur when chemical or enzymatic catalyst or buffering agent is used^{151,157,158}. Continuous scission causes higher rate of monomer conversion. In thick and bulky implants, degradation is disorderly between surface and core portion due to surrounding tissue. Adhesion of tissue on surface results in high rate of degradation. Fine surface and/or less porosity have a better tendency to withstand acid break down and reduce hydrolysis¹⁵⁹⁻¹⁶¹. Molecular construction (R-group), degree of crystallinity and micro structure intervene in degradation rate. Amorphous region retains higher amount of water due to high free volume and more segmental mobility, thereby leading to fast hydrolytic degradation compared to crystalline region. Crystalline region can act as a reinforcement, which increases degradation time and mechanical properties¹⁶²⁻¹⁶⁶.

Self-reinforced homopolymer of PLA shows higher strength and high crystallinity. It takes over five years for complete absorption and amorphous poly (D, L-Lactide) can absorb in 1.5 years^{167,168}. Copolymer of PGA-PLA shows faster degradation than homopolymer of PLA. PDO and PGA degrade around two months and complete reduction of molar mass occurs in 9-12 months¹⁶⁹. R-group of PGA is more sensitive towards water. By varying composition of PLA and PGA, degradation rate can be specified without rapid

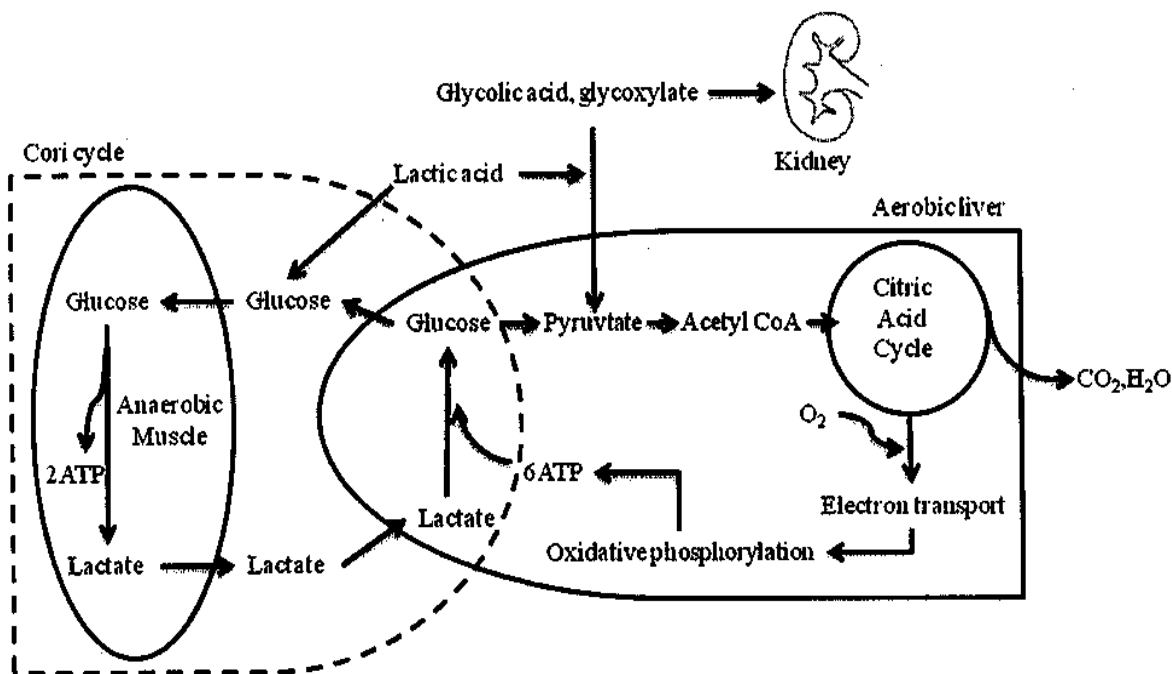


Fig. 1— Bio-absorption of monomers in metabolism

degradation and control release of acidic breakdown products. Hydrolysis of PLA, PGA and PDO breaks down into lactic acid, glycolic acid and glyoxylate respectively. Glycolic acid and glyoxylate are excreted via kidney or converted as pyruvate and further conversion into acetyl CoA involving citric acid cycle. Lactic acid can be converted into glucose, which further breaks down into adenosine triphosphate (ATP) via cori cycle (energy cycle) (Fig. 1)¹⁷⁰⁻¹⁷².

Applications

Many polymer devices commercially available are most commonly smart screws, smart pins (SR-LPLA-Bionx Implants), phantom softthread soft tissue fixation screw (LPLA-DePuy), orthosorb pin (PDO-J and J Orthopedics), biologically quiet staple (85:15/DL-PLGA-Instrument Makar), meniscal stinger (LPLA-Linvatec) and bio-anchor (LPLA-Linvatec). Denti *et al*¹⁷³ investigated effect of four different bioabsorbable interference screws for bone block fixation. PLLA screw (Linvatec), PGA screw (Smith & Nephew), PLA-co-PGA screw (Instrument Makar), PLA screw (Arthrex) are available commercially and reported that PGA screws are absorbed more quickly than PLA.

Orthopaedic Application

Bio-absorbable implants in orthopaedic surgery have mainly been used to eliminate implant removal operations¹⁷⁴⁻¹⁷⁶. Self reinforced PLA and PGA screws and rods have excellent strength for bone fixation. PLA (molar mass, 1,00,000) has excellent mechanical property for implantation and other applications. PGA (molar mass, 20,000-1,45,000) is used for orthopaedic implants^{177,178}. To a large extent with orthopaedic and traumatologic implants involved, incidence of infection¹⁷⁹ in conjunction with bio-absorbable osteosynthesis devices ranges from 3-4%. To facilitate tissue regeneration and uniform load transformation, scaffolds are used as a temporary support to body structures. Reconstruction of finger joints was performed using scaffolds folded of VicrylR (co-polymer of 90% PGA and 10% PLLA) or EthisorbR (made up of PDS and co-polymer of 90% PGA and 10% PLLA) fleeces¹⁸⁰. Recently, electrospun nanofiber scaffolds showed greater promise and potential for many biomedical applications [tissue engineering, wound dressing, immobilized enzymes and controlled-delivery of drugs (genes)].

Tissue Engineering

In tissue engineering, PDO scaffolds performed with high flexibility, slow degradation, higher strength retention and lower inflammatory response rates when compared to Vicryl (glycolic-co-lactic acid polymer) and Dexon (polyglycolic acid). Monofilament suture typically loses 50% of its initial breaking strength after 3 weeks^{181,182}. Degradation time of PLA-PGA scaffolds depends upon the ratio of monomers in copolymer composition. Optimum pores for bone regeneration, regeneration of skin and in growth of cells are 100-150, 20-125 and 20-60 respectively¹⁸³. Surface modification of electrospun scaffolds with suitable bioactive molecules can effectively increase biological activity, useful for specific biomedical applications¹⁸⁴. Polyheterocycle based nanoparticles-PLA composites are observed as a biocompatible conducting material, which can be used as a biosensor for tissue engineering^{185,186}. *In vivo* study suggested that polyheterocycles would have slow degradation and renal clearance^{187,188}.

Control Drug Delivery Systems

Control drug release can occur by desorption of surface bound drug and/or diffusion through nanoparticle matrix. Biodegradable materials are intended to degrade within the body after release of active agent has been completed. This eliminates need for removal of material after drug release¹⁸⁹⁻¹⁹³. BAP's can be tailor-made to achieve both controlled drug release and disease specific localization by tuning polymer characteristics by varying copolymer ratio to crystallinity can control via drug delivery¹⁹⁴⁻²¹⁰. PLGA (Poly(lactic-co-glycolic acid)) of different molecular weights (10-100 kDa) having different copolymer ratios (50:50, 75:25 and 85:15) are available in the market. Molecular weight and copolymer molar ratios influence degradation process and release profile of the drug entrapped^{211,212}. Nanoparticles with high crystallinity could lead to the formation of a microchannel structure and a large surface area in polymer matrix, making drug release easily²¹³.

Storage and Sterilization

Implanting bio-absorbable device should be sterilized by γ -radiation or ethylene oxide (EtO) or plasma etching before clinical use²¹²⁻²¹⁴. Molecular weight, mechanical property and performance of polymers are also dependent on sterilization and storage conditions. EtO sterilization residual gas causes toxic effect in body and hence not a preferable method for bio-implants²¹⁵. γ -radiation is preferable method for bio-implants; low dose of 250 kGy γ -radiation enhances chain scission in

polymer and higher dose introduce crosslink's between linear chains²¹⁶⁻²¹⁸. Bio-absorbable materials are stored in low temperature, dry environment and low humidity²¹⁹.

Conclusions

Use of BAP's has marvelous growth in a variety of biomedical applications. Polymer implantation devices are convenient to design structure and tailor-made property according to environment and usage conditions. Replacements of heavy metal implants by BAPs have been successively carried out. For material development, non-toxic catalyst with simple polymerization method needed. For optimum performance, BAP's should be sterilized and preserved under ideal conditions.

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